

Summer 6-26-2005

X. Oxidation of silicon: Role of Si/SiO₂ interface in optical and electrical properties

M Tomozawa

Rensselaer Polytechnic Institute

Follow this and additional works at: <https://preserve.lehigh.edu/imi-tll-conferences-17thuniversityglassconference>

Part of the [Materials Science and Engineering Commons](#)

Recommended Citation

Tomozawa, M, "X. Oxidation of silicon: Role of Si/SiO₂ interface in optical and electrical properties" (2005). *17th University Glass Conference*. 9.

<https://preserve.lehigh.edu/imi-tll-conferences-17thuniversityglassconference/9>

This Video is brought to you for free and open access by the Glass Conferences and Workshops at Lehigh Preserve. It has been accepted for inclusion in 17th University Glass Conference by an authorized administrator of Lehigh Preserve. For more information, please contact preserve@lehigh.edu.



An IMI Video Reproduction of Invited Lectures
from the 17th University Glass Conference

OXIDATION OF Si: Role of Si/SiO₂ INTERFACE

M. Tomozawa and S.-R. Ryu

Materials Science and Engineering
Department

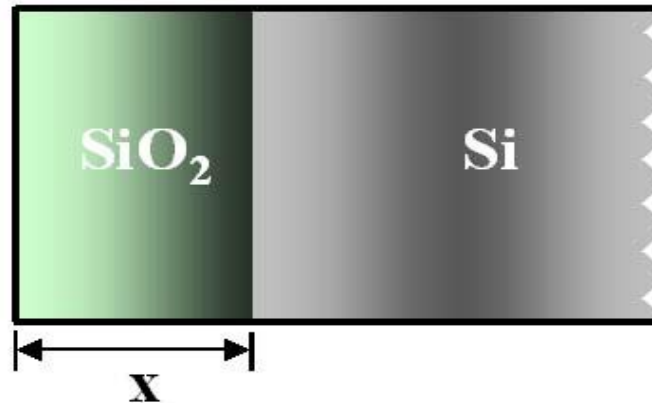
Rensselaer Polytechnic Institute

Troy, NY 12180-3590 USA

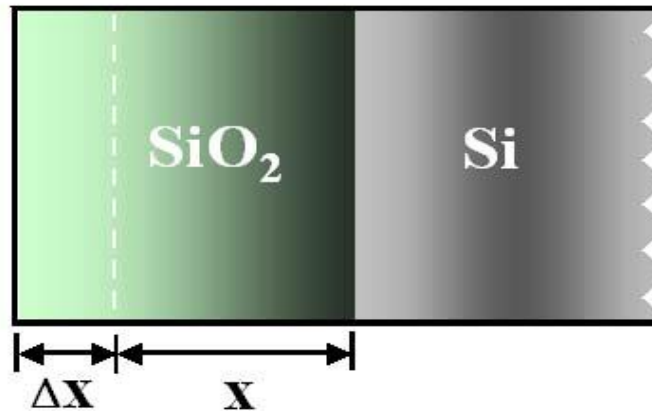


Oxidation of Silicon

Heat treated
time t



Heat treated
time $t + \Delta t$



Oxidation of Silicon

Si is oxidized to form amorphous SiO₂ at high temperature in dry (O₂) atmosphere or wet (H₂O) atmosphere.

The oxidation kinetics follows the linear-parabolic law at a constant temperature.

$$x_0^2 + Ax_0 = B(t + \tau)$$

where x_0 : SiO₂ film thickness

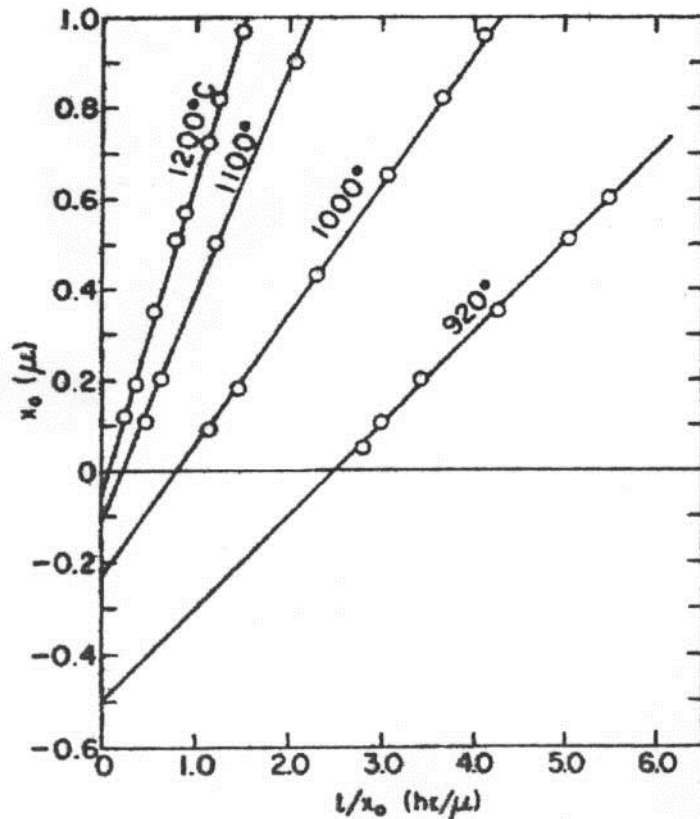
t : heat-treatment time; τ : off-set time

B : parabolic rate constant; B/A : linear rate constant

Deal and Grove, J. Appl. 36 (1965) 3770.



Oxidation of Silicon



$$X_0 = -A + Bt/x_0$$

Wet (95°C H₂O)
oxidation, $\tau = 0$

Deal and Grove



Oxidation of Silicon

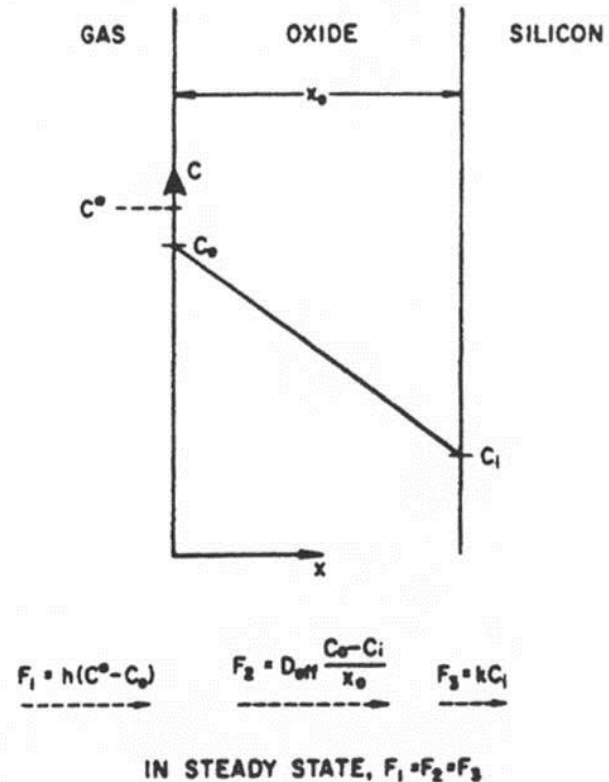
Traditionally

Linear growth is attributed to the interface reaction-controlled process.

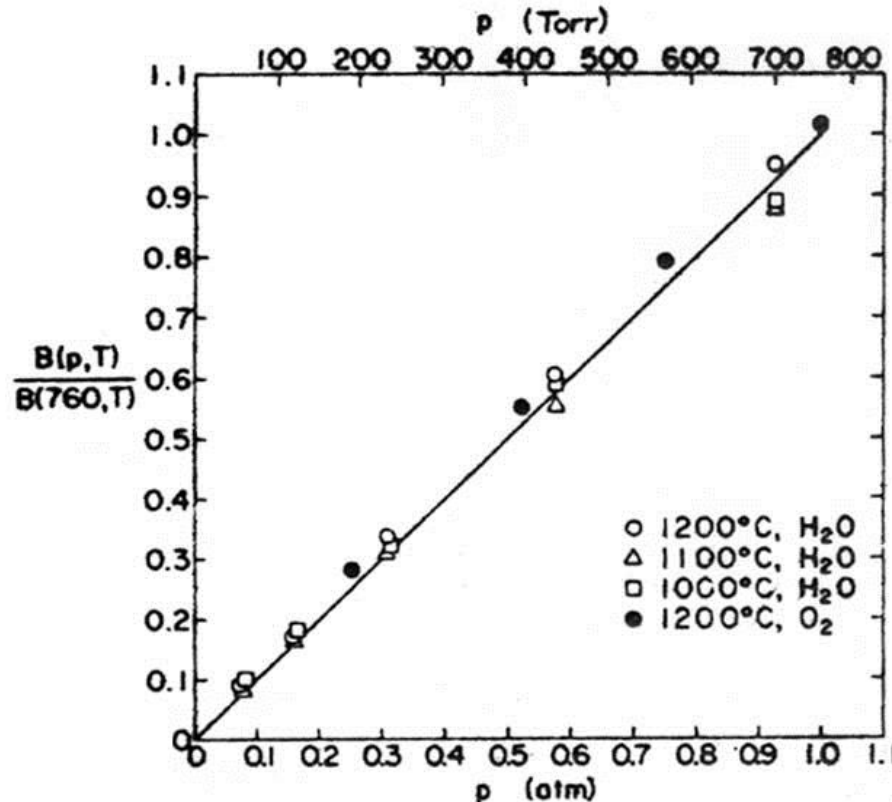
Parabolic growth is attributed to diffusion-controlled process.

Diffusing species are H_2O in wet and O_2 in dry.

B.F. Deal and A.S. Grove, J. Appl. Phys. 36, 3770 (1965).



Oxidant pressure effect



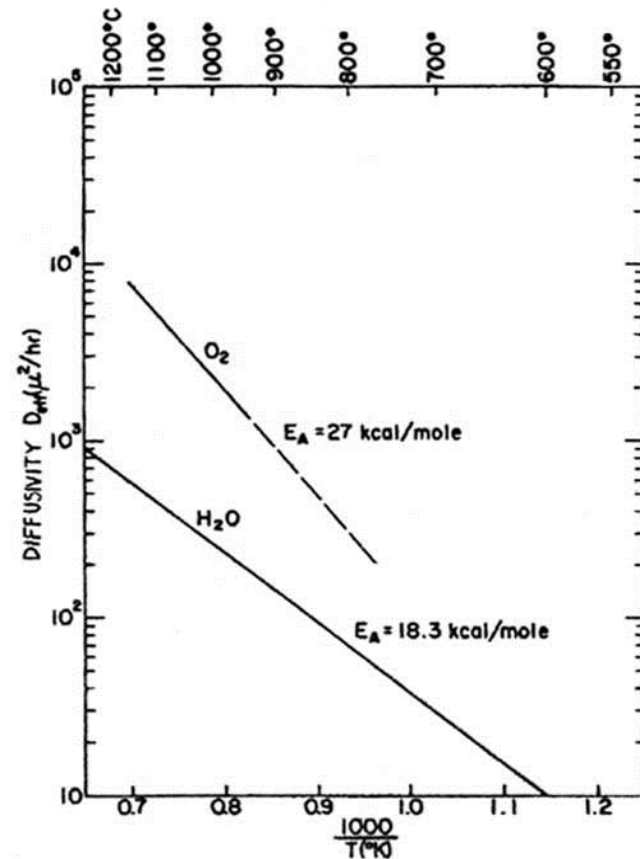
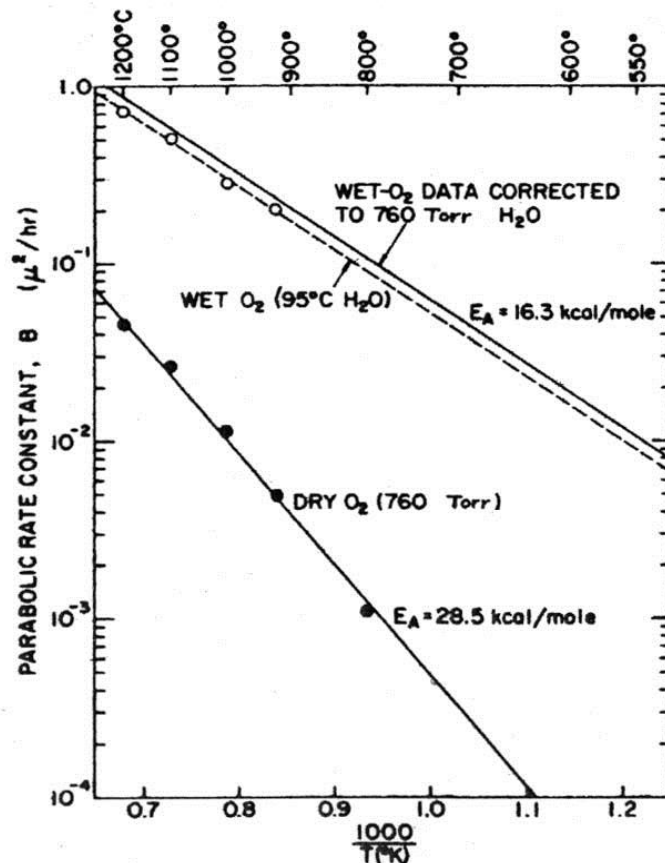
Parabolic rate constant, B , is proportional to vapor pressure.

Linear rate constant B/A is proportional to vapor pressure. (Deal and Grove); or proportional to square root of vapor pressure (Deal, Hess, Plummer and Ho, J. Electrochem. Soc., 125, 339)



Parabolic rate constant

$B = 2DC^*/N$; N: number of oxidant in a unit volume of SiO_2 (Deal and Grove) and D: Diffusion coefficients



Anomalous phenomena related to Si/SiO₂

Wet oxidation is faster than dry oxidation even though the diffusion coefficient of water appears smaller than that of oxygen.

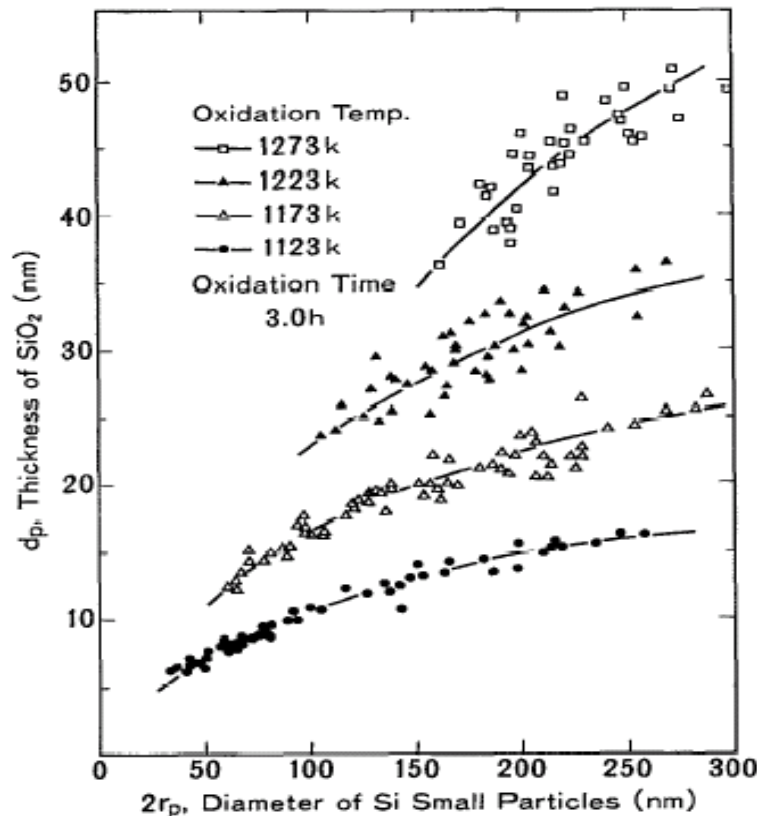
Origin of linear rate constant—reaction rate of Si with oxidant is fast. (Doremus, JAP, 66 (1989)4441; Mott, Rigo and Rochet, Philo. Mag. B, 60 (1989) 189; Bongiorno and Pasquarello, Phys. Rev. Letters, 93 (2004) 086102).

Oxidation rate is slower for nano-Si.

Oxidation rate depends upon the orientation of Si crystal.



Oxidation of nano-silicon

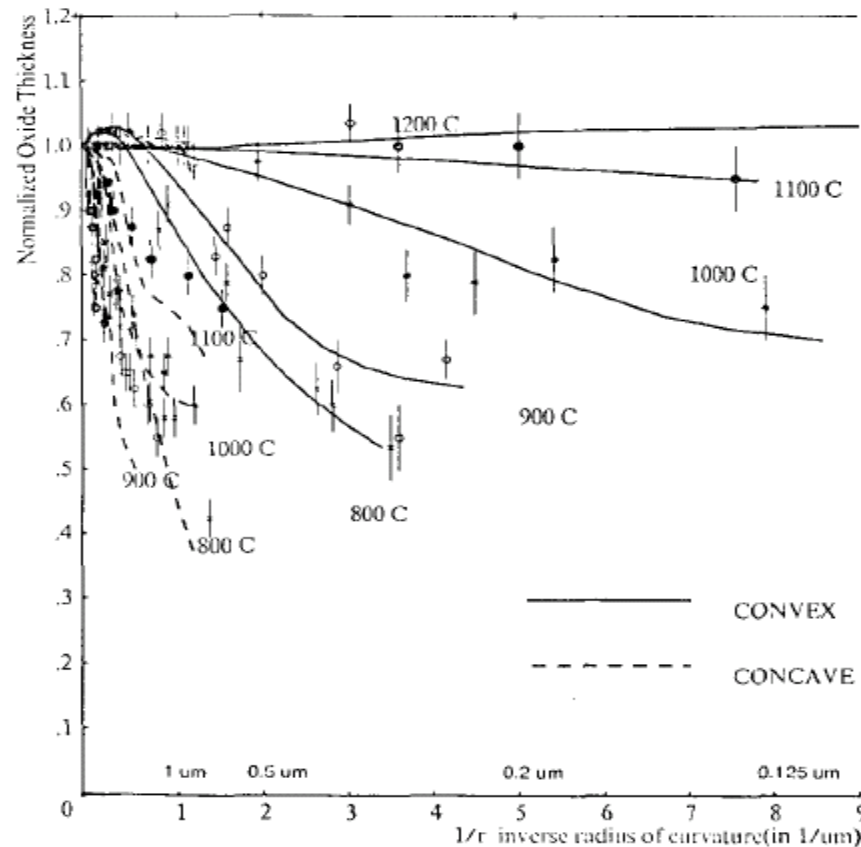


SiO_2 film growth rate is slower for smaller silicon particles

R. Okada and S. Iijima,
Appl. Phys. Letters, 58,
1662 (1991).



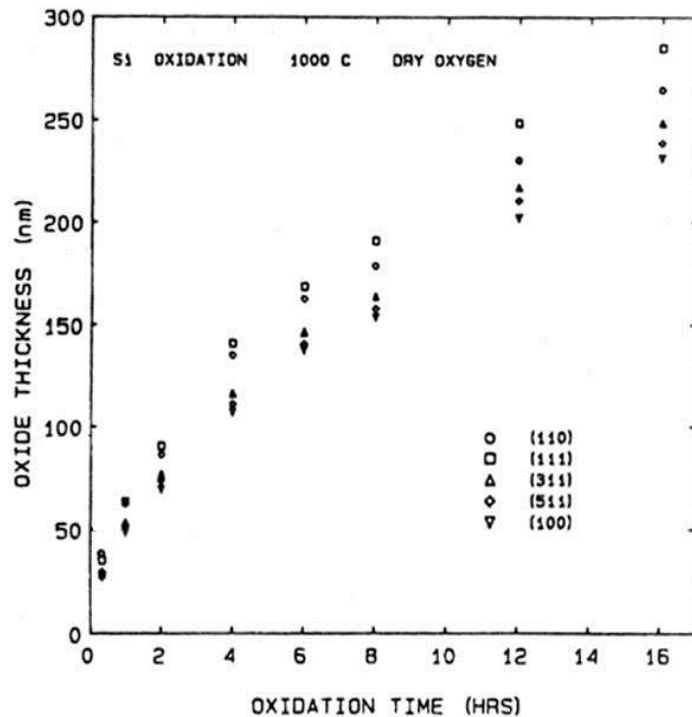
Oxidation rate of Si with curved surface



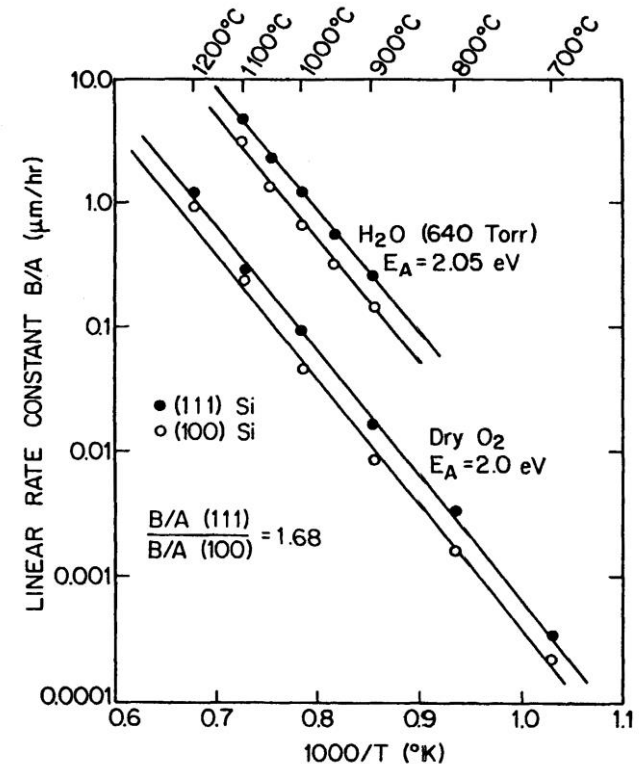
Kao, McVittie, Nix, and Saraswat, IEEE
ED-35, 25(1988).



Effect of Si crystal orientation on oxidation rate



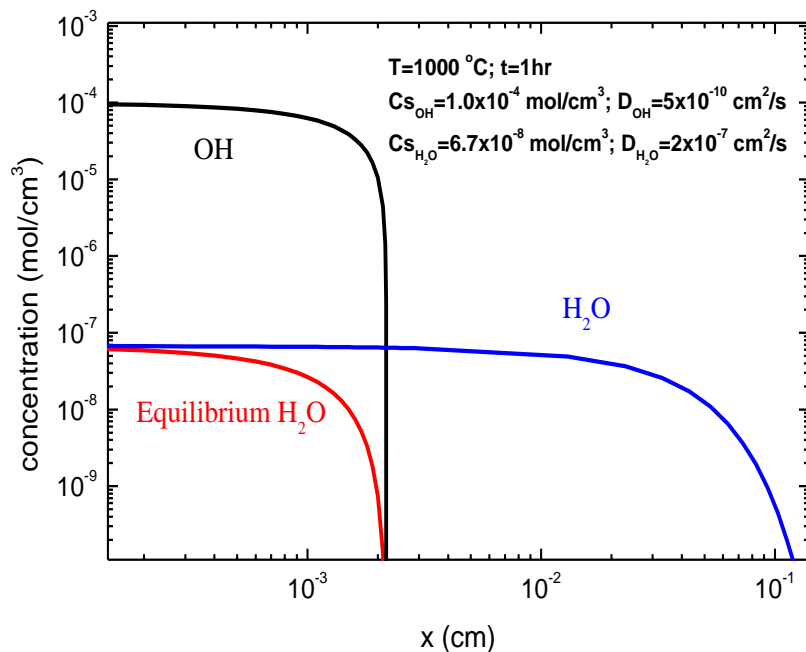
Lewis and Irene
J. Electrochem. Soc., 134
(1987) 2332. Oxidation rate:
(111)>(110)>(100)



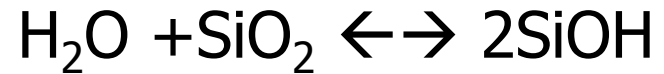
Deal, J. Electrochem. Soc. 125
(1978) 576.



Expected water (H₂O and OH) diffusion profiles



Water diffusion, 1000°C,
355 Torr water vapor, 1 h.



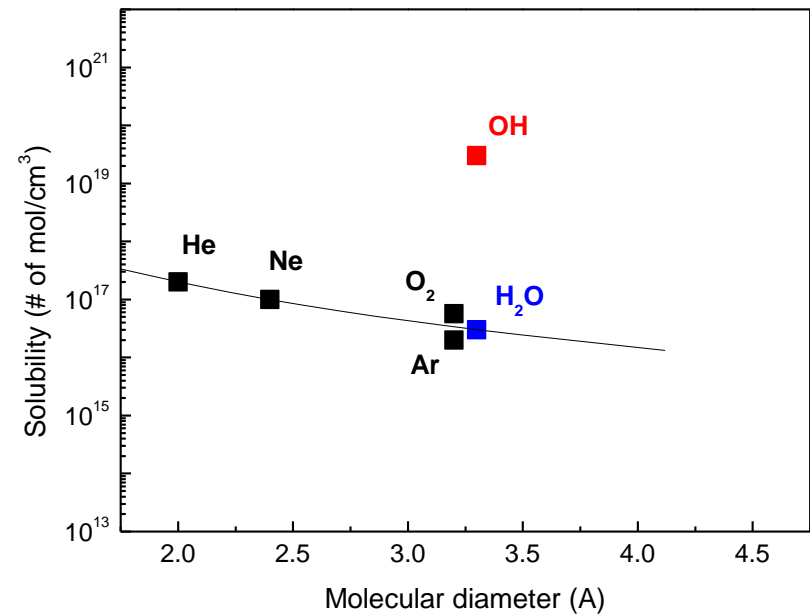
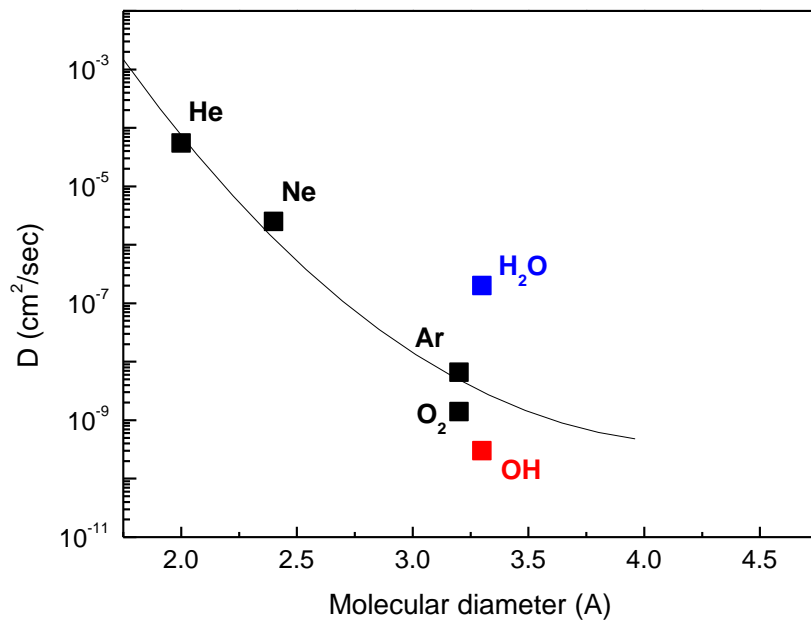
H₂O diffuses and reacts
with SiO₂ to form SiOH.

If the reaction equilibrium
holds strictly, H₂O and OH
(solid line) should have
same depth of diffusion.

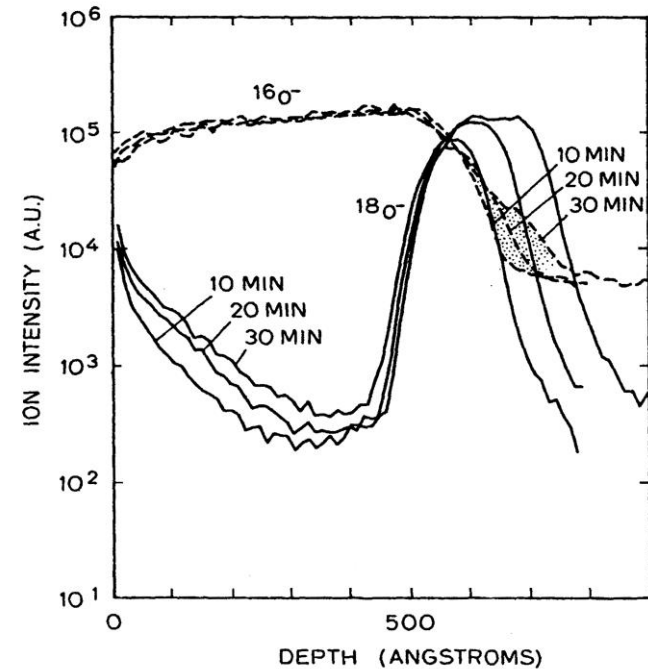
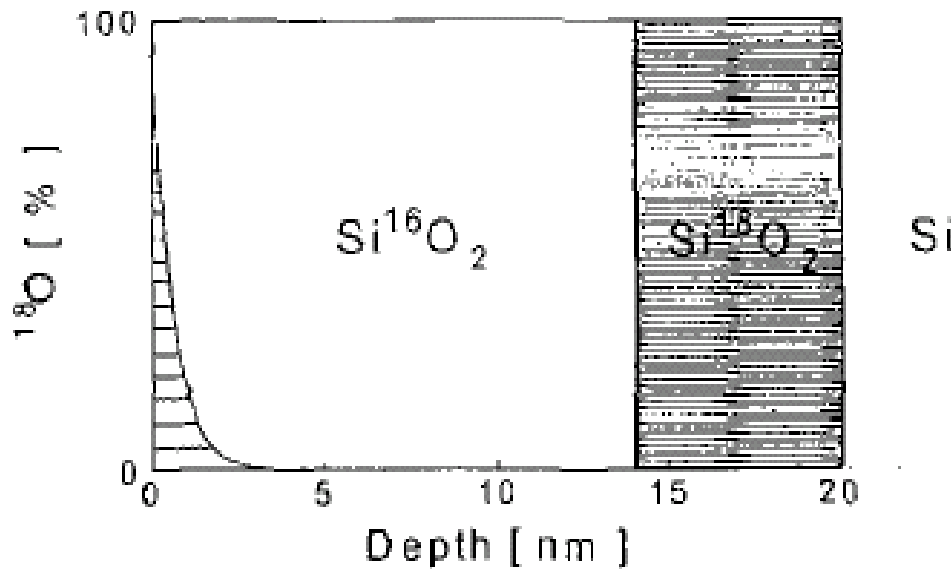
If H₂O is not restricted by
the equilibrium, it can
diffuse deeper.



Diffusion coefficient and solubility



$^{18}\text{O}_2$ diffusion during Si oxidation



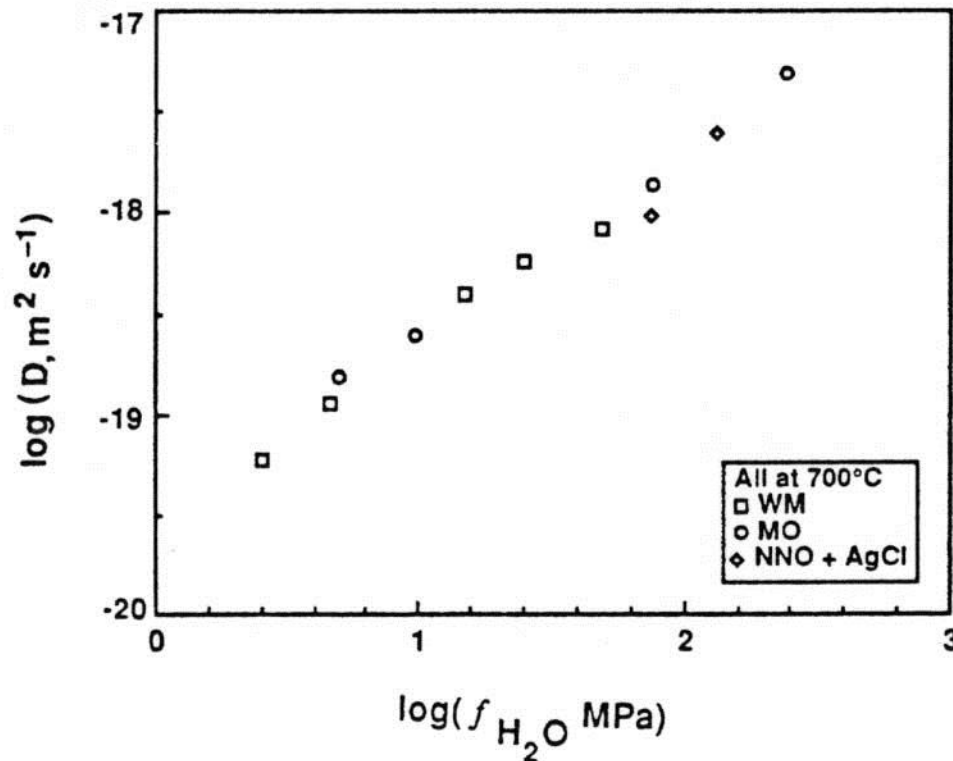
Stedile, Baumvol, Oppenheim, Trimaille,
Ganem and Rigo, Nucl. Inst. Method in
Phys. Res. B 118 (1996) 493.

Treatment in $^{16}\text{O}_2$ Followed by $^{18}\text{O}_2$

Han and Helms, JAP, 59
(1986)1768.



Apparent diffusion coefficient of ^{18}O as a function of water vapor pressure

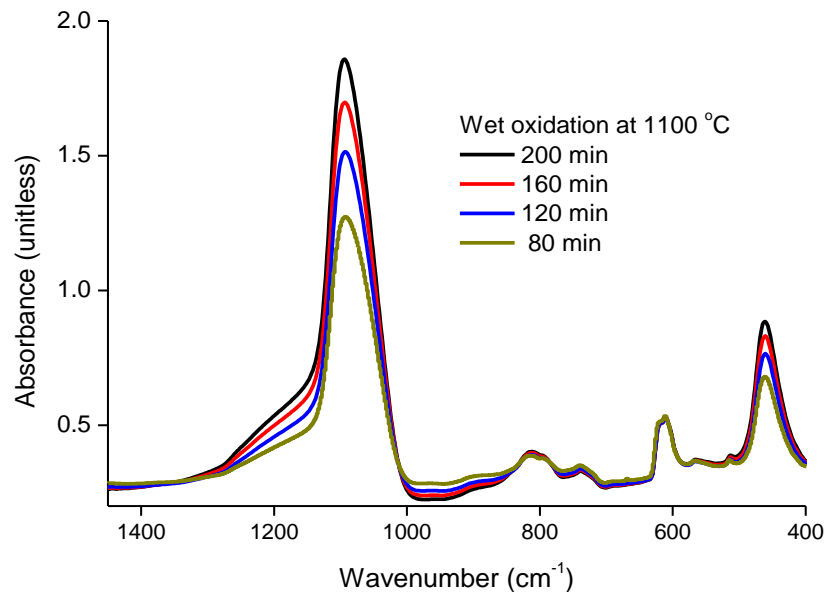


Farver and Yund,
Chem. Geology, 90
(1991) 55. Quartz:

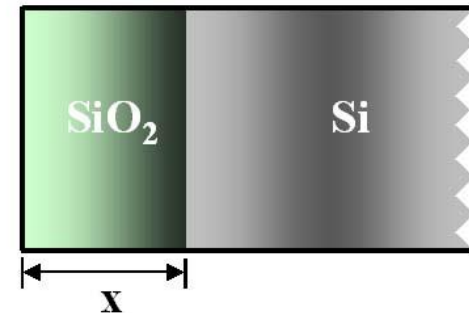
Similar analysis was
made for SiO_2 film
By Doremus in Diffusion
of Reactive Molecules in
Solids and Melts, Wiley
(2002).



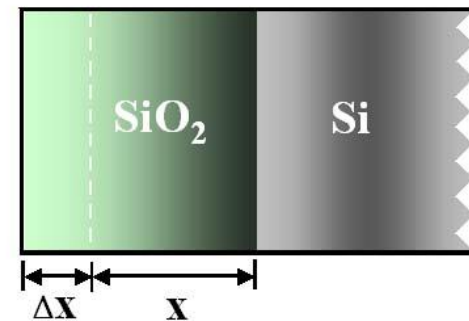
IR absorption of amorphous SiO₂ film



Heat treated
time t



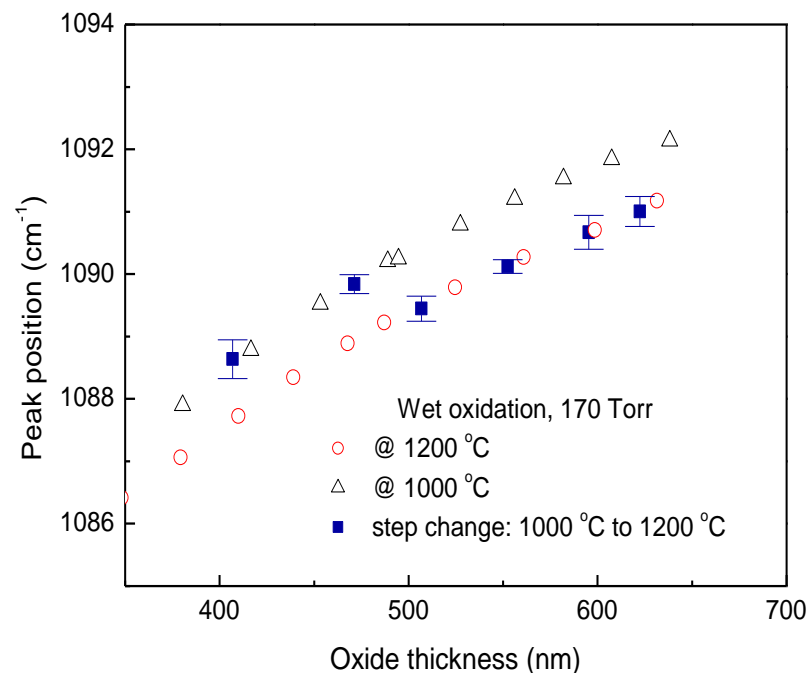
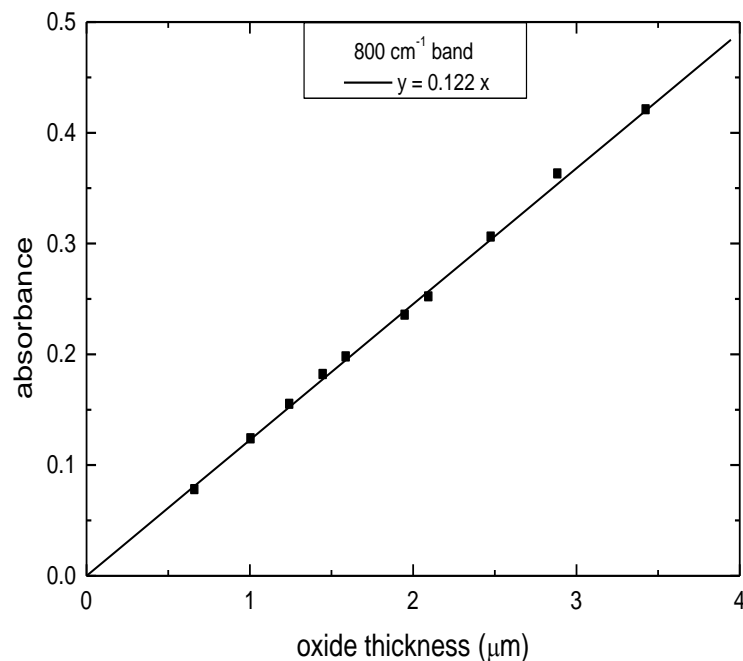
Heat treated
time $t + \Delta t$



Absorbance of IR SiO₂ structural bands can be used to determine SiO₂ film thickness.



IR peak wavenumber silica structural band and thickness of SiO₂ films



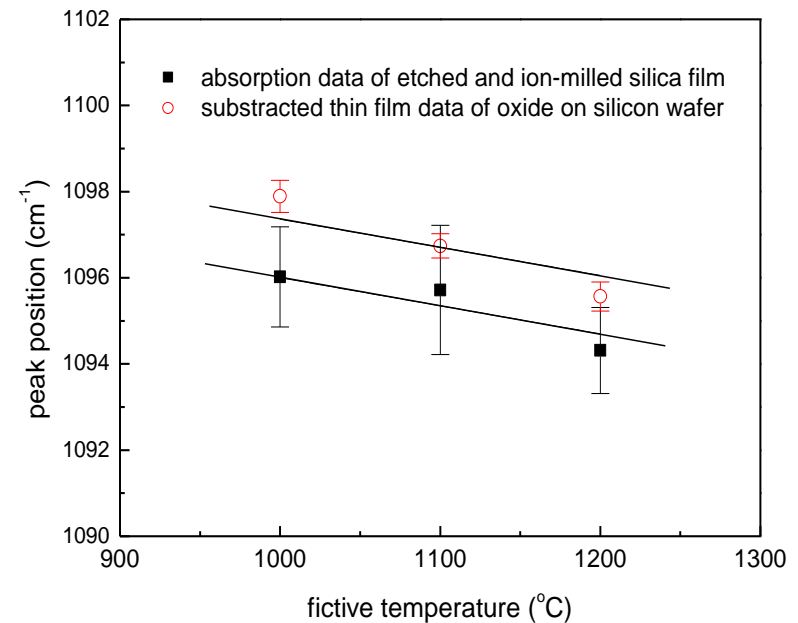
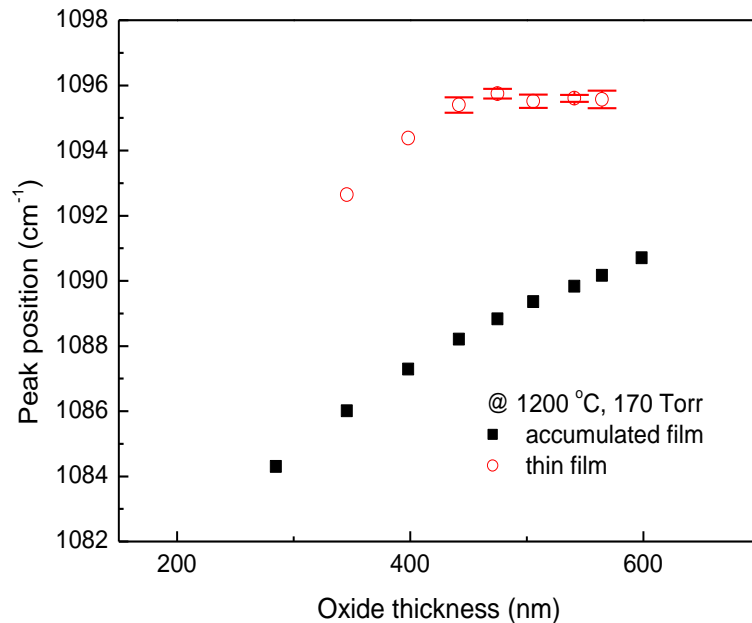
Peak wavenumber of IR SiO₂ structural band

IR Peak wavenumber can change with

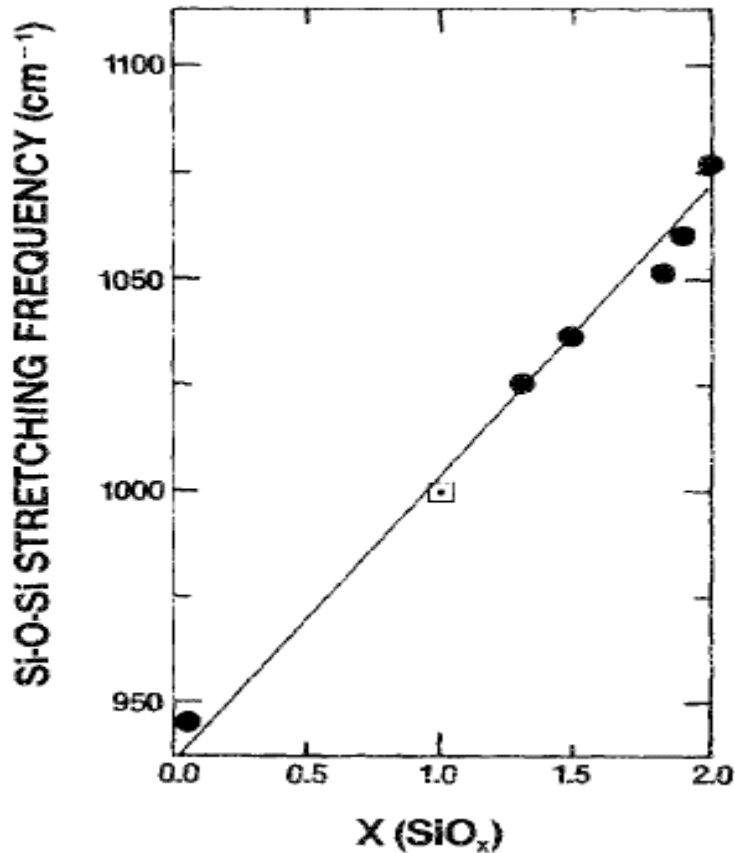
1. SiO₂ fictive temperature—small effect
~0.006 cm⁻¹ /100°C change of T_f at 1200 cm⁻¹ peak.
2. Stress (density change)-intermediate effect
<1 cm⁻¹/GPa
91 cm⁻¹ shift for 1 g/cm³ density change.
High density of 2.4 g/cm³ for 1.5 nm film was reported. 18 cm⁻¹ for 0.2 g/cm³ change.
3. Non-stoichiometry-large effect
~50 cm⁻¹ decrease for x= 2 → 1.5 in SiO_x



Fictive temperature effect on IR silica structural band peak wavenumber



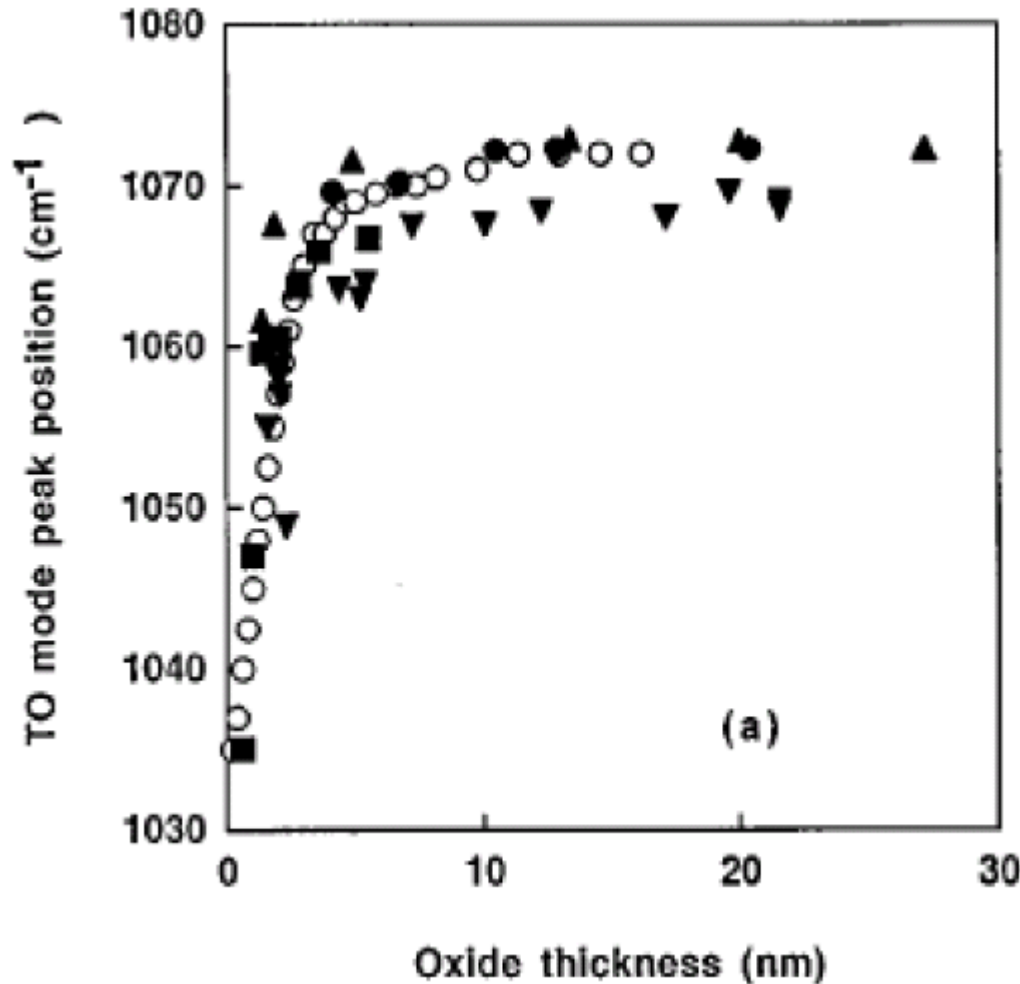
Effect of non-stoichiometry on IR silica structural band peak wavenumber



P.G. Pai, S.S. Chao, Y. Takagi,
and G. Lucovsky, J. Vacuum
Sci. Technol. 4, 689 (1986).



IR peak wavenumber at Si/SiO₂ interface

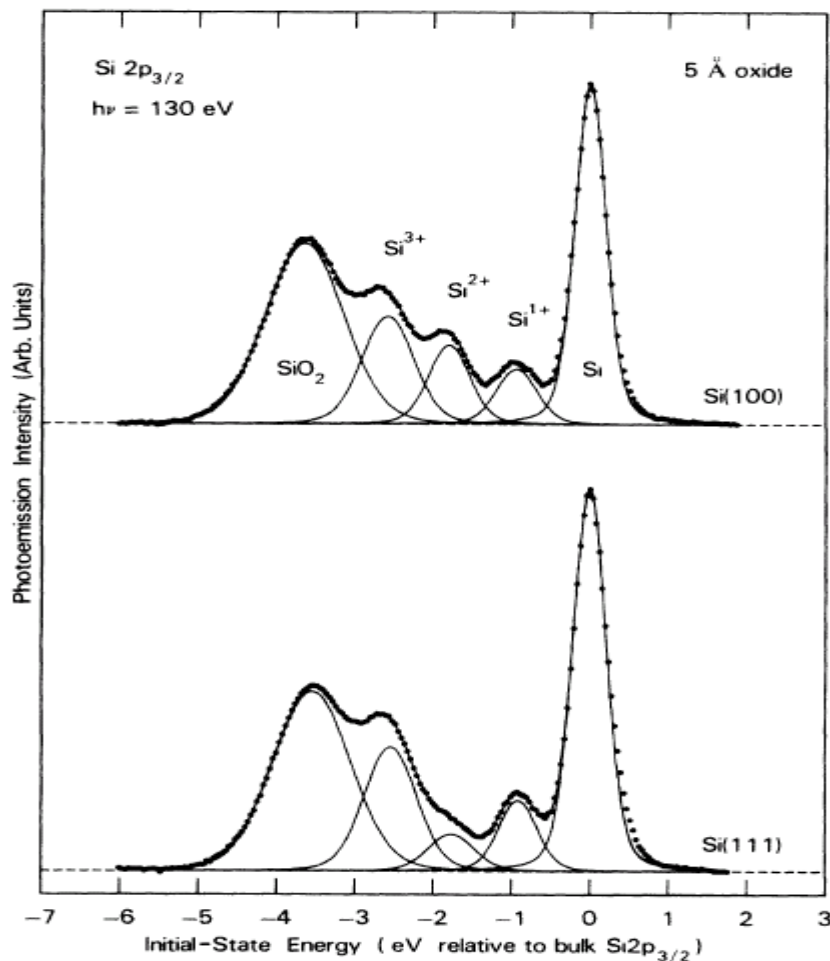


R.A.B. Devine, Appl. Phys. Lett., 68, 3108 (1996).

At the interface the composition is estimated to be **SiO_{1.45}**



Si/SiO₂ interface structure

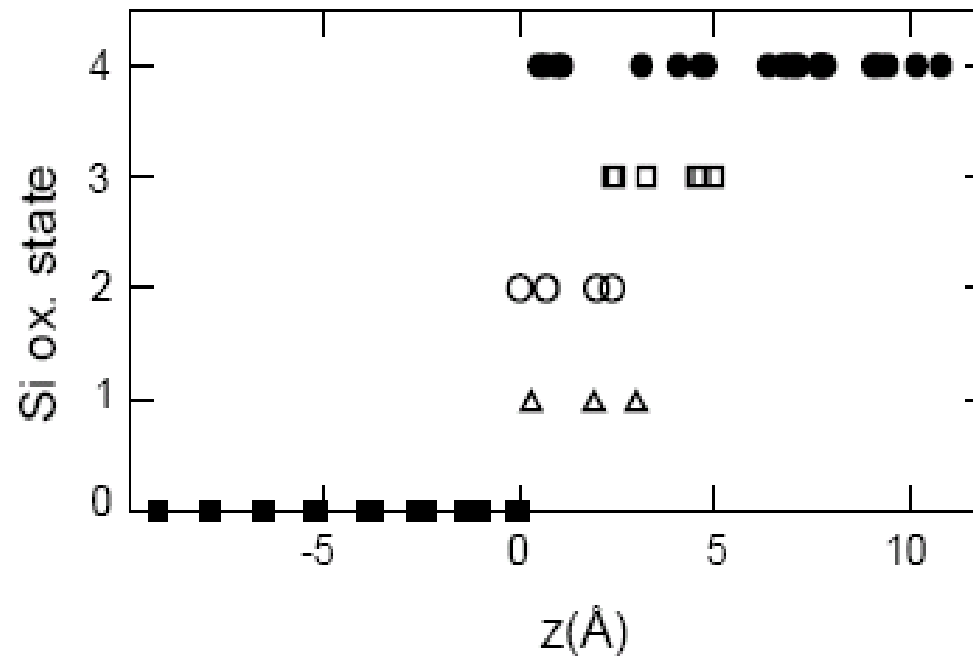


Himpsel, McFeely,
Taleb-Ibrahimi and
Yarmoff, Phys. Rev. B
38 (1988) 6084.

Core-level
spectroscopy with
Synchrotron radiation



Si/SiO₂ interface structure



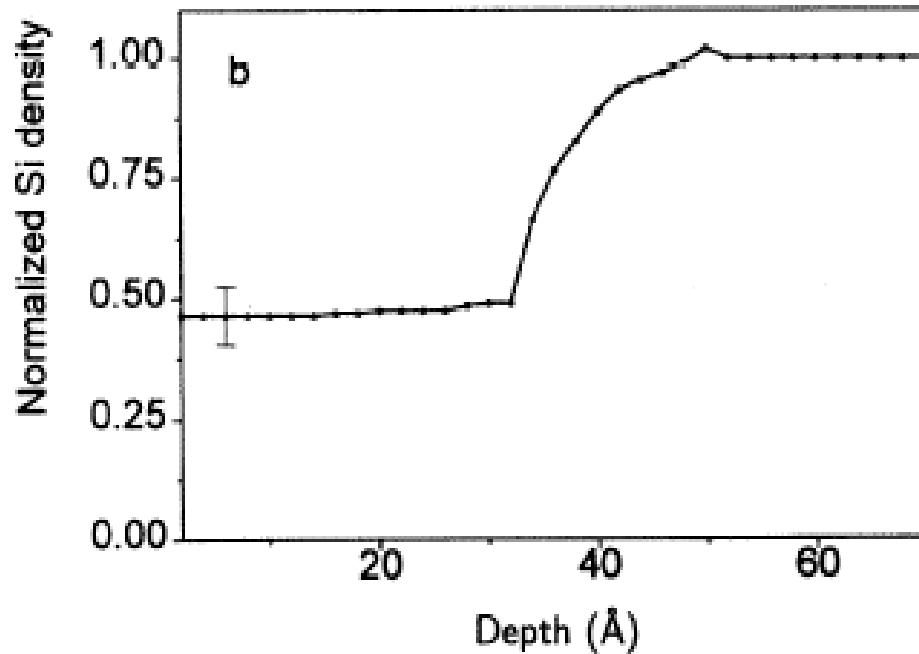
Bongiorno and Pasquarello, Appl. Surf. Sci. 234 (2004) 190.

Atomistic model based upon experimental data

Three atomic layers average SiO_{1.46} (exp. SiO_{1.70}, SiO_{1.43})

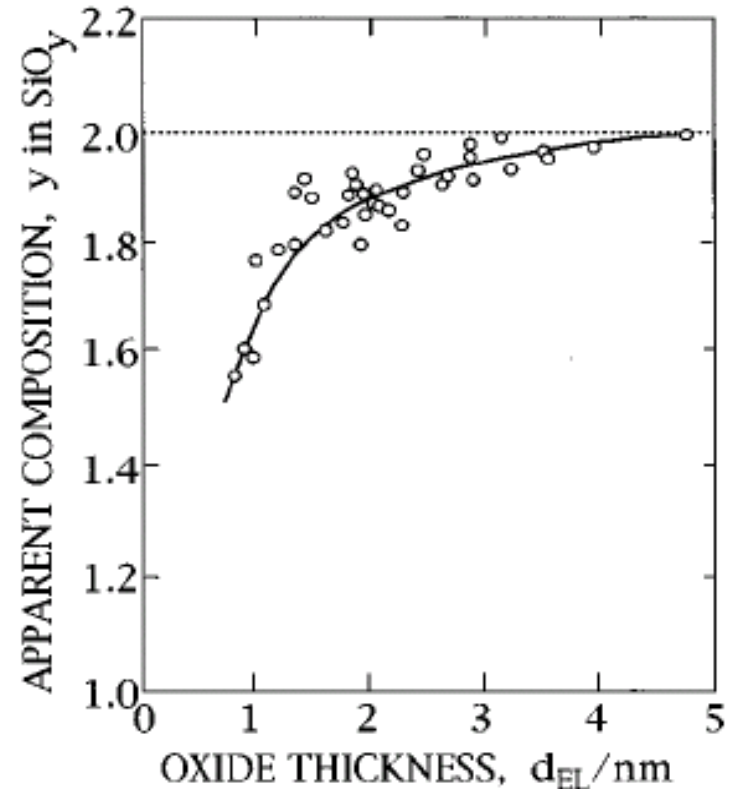


Si/SiO₂ interface



Gusev, Lu, Gustafsson, and
Garfunkel, Phys. Rev. B
52 (1995) 1753.

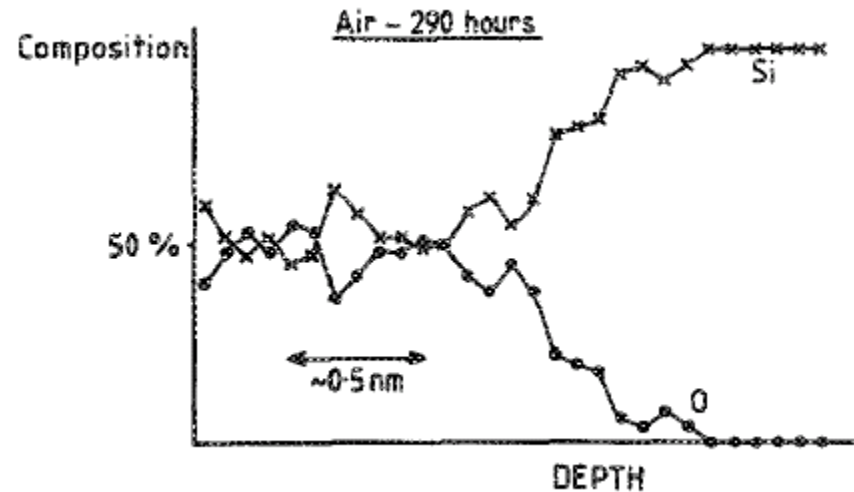
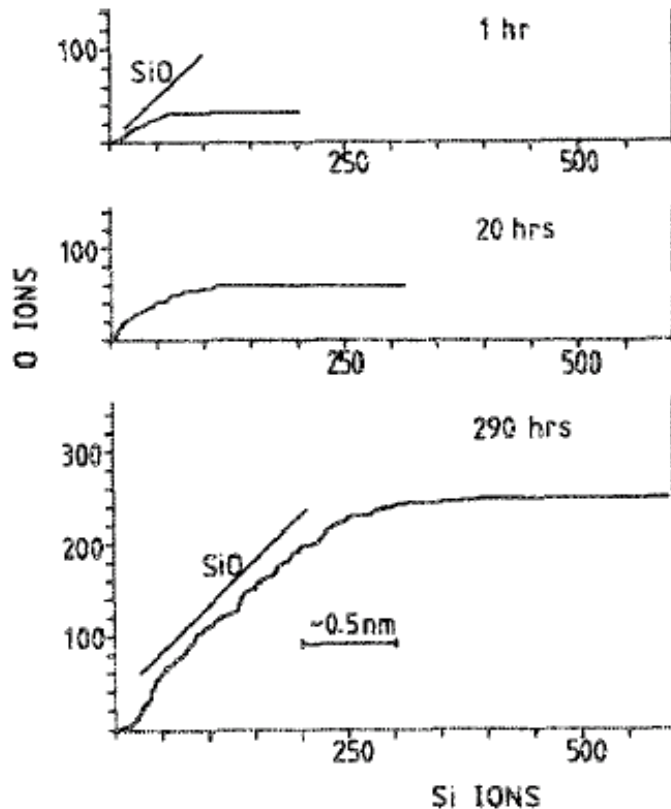
From medium-energy ion scattering



Iwata and Ishizaka, J. Appl.
Phys. 79 (1996) 6653. (from
ESCA data)



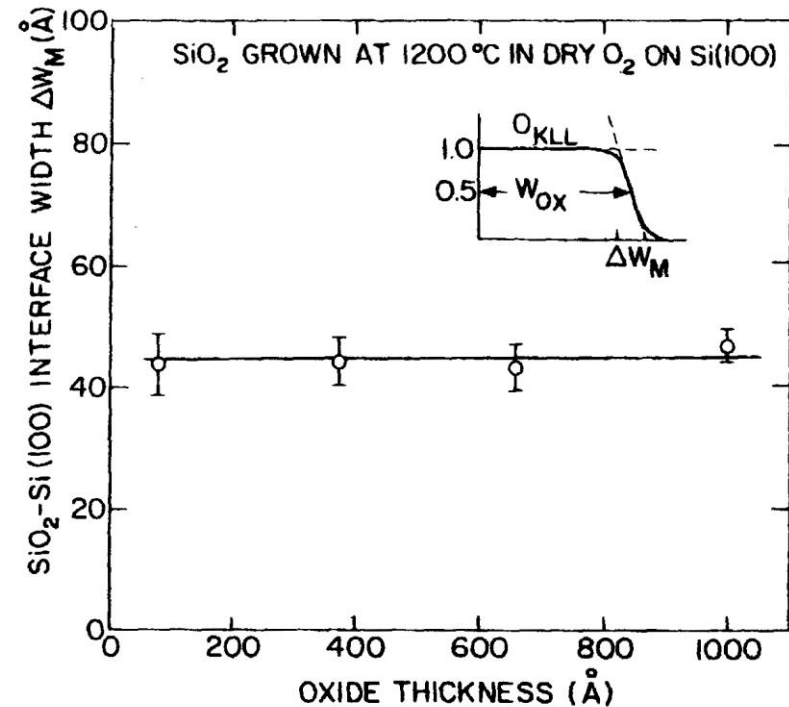
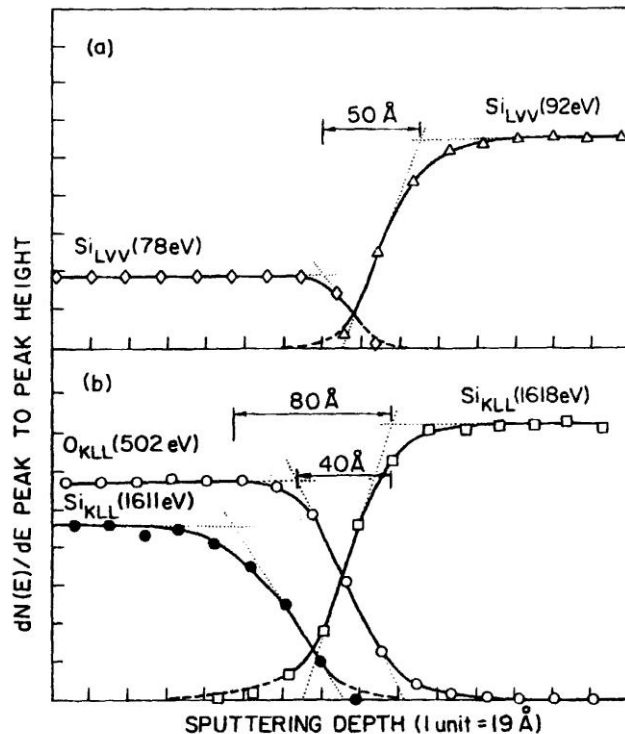
Si/SiO₂ interface structure



Grovenor and Cerezo, J. Appl. Phys. 65 (1989) 5089.
Room temperature air, after HF cleaning.
Pulse laser atom probe microanalysis.



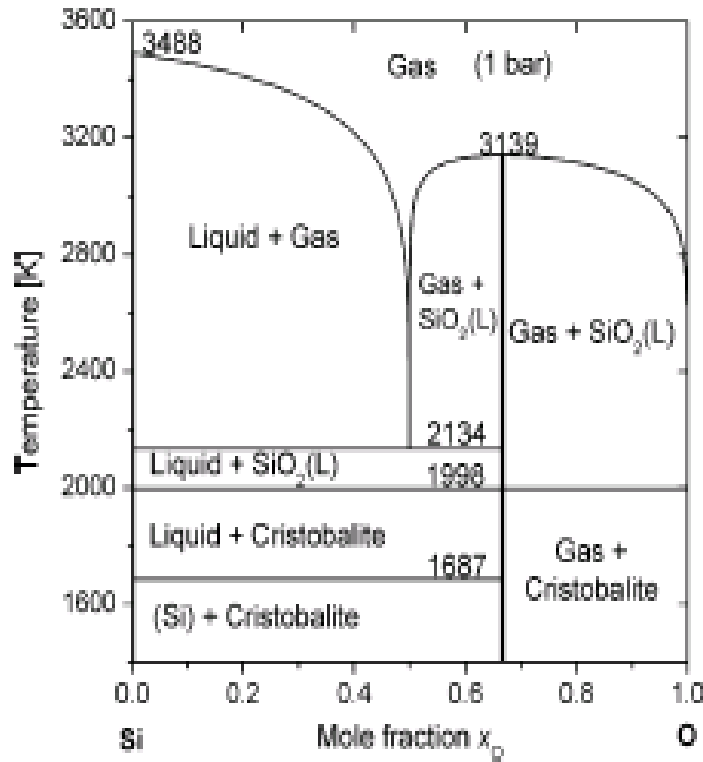
Si/SiO₂ interface structure



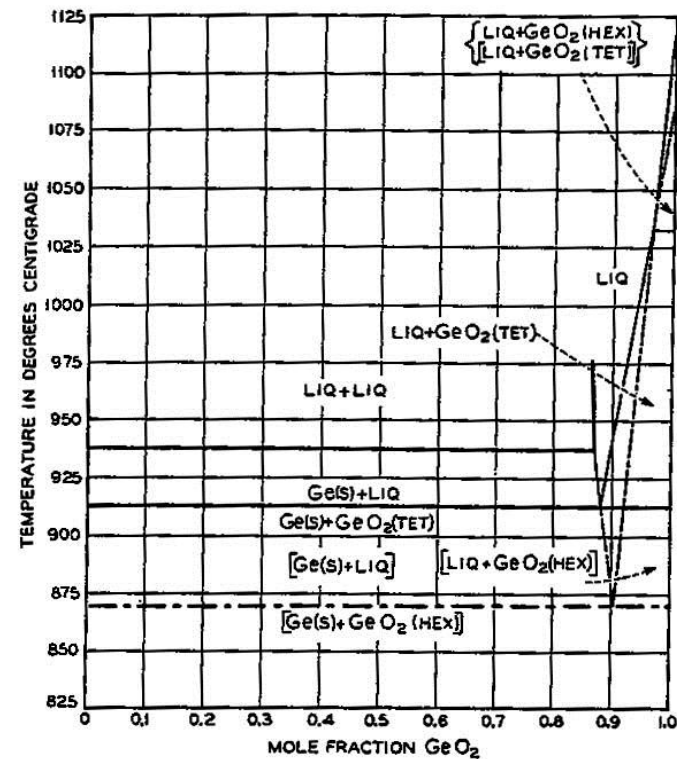
Johannessen and Spicer, and Spicer, JAP, 47 (1976) 3028.
Auger Analysis.



Phase separation of Si/SiO₂ (Ge/GeO₂) system



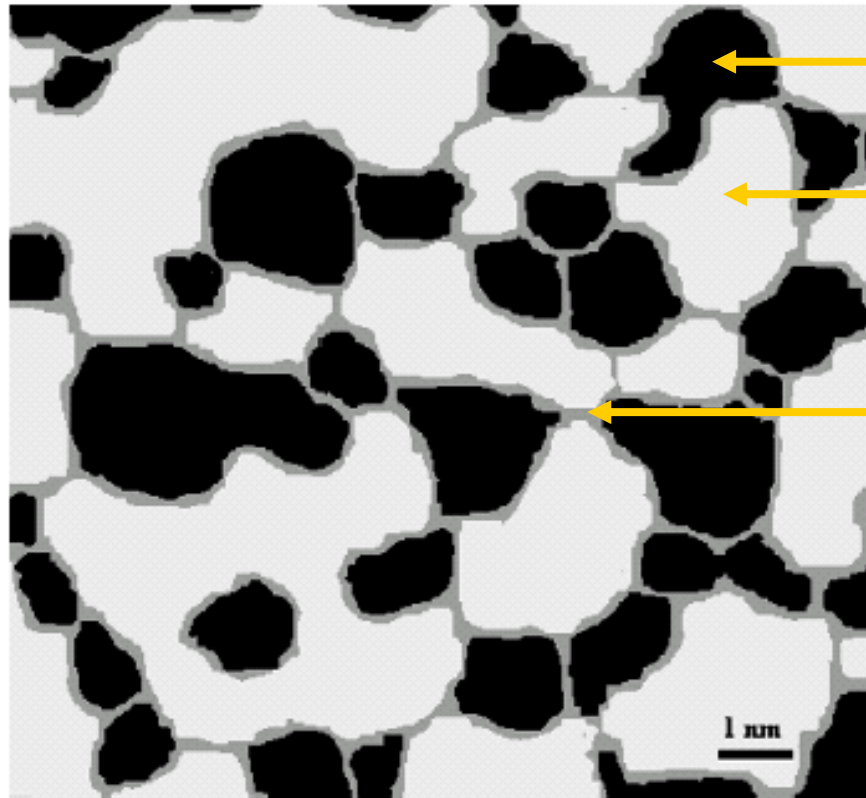
Schmure et al., J. Non-Cryst. Solids, 336 (2004) 1.



Trumbone et al. J. Chem. Phys. 24 (1956) 1112.



Phase Separation of a-SiO



Black: Si

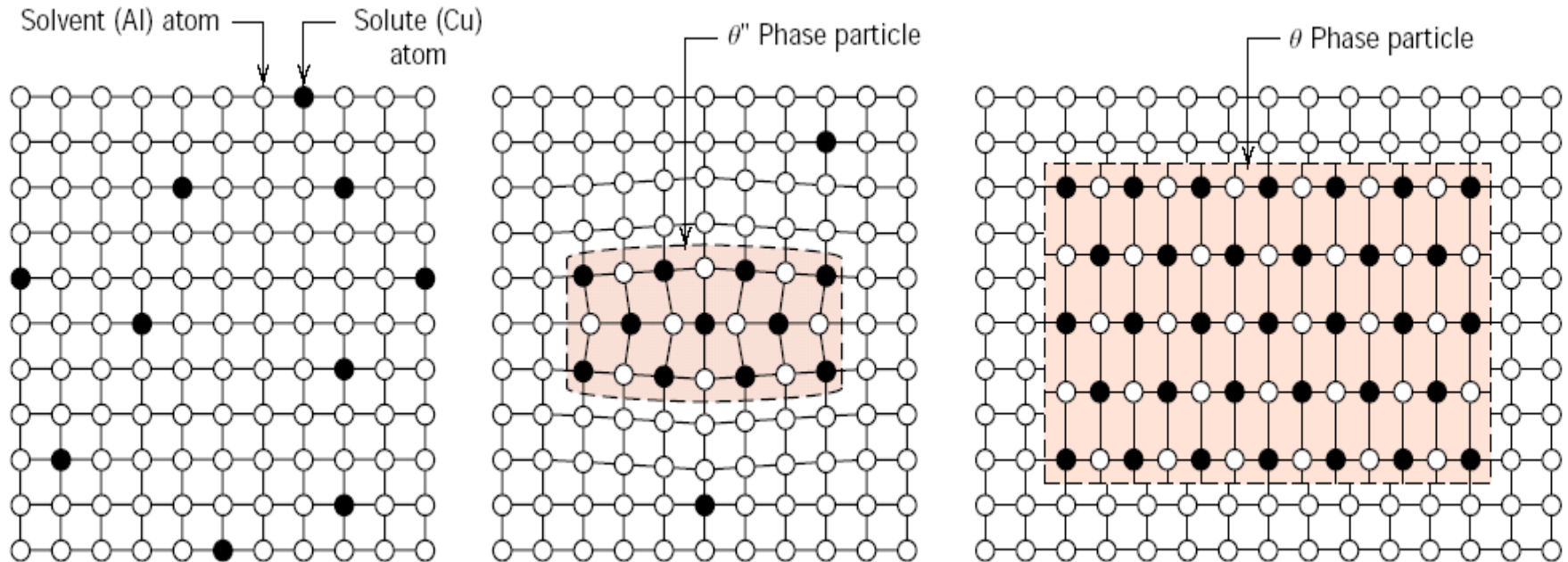
Light Gray: SiO₂

Gray: Suboxide, SiO_x

Hohl, Wieder, van Aken, Weirich, Denninger, Vidal, Oswald, Deneke, Mayer and Fuess, J. Non-Cryst. Solids, 320 (2003) 255.



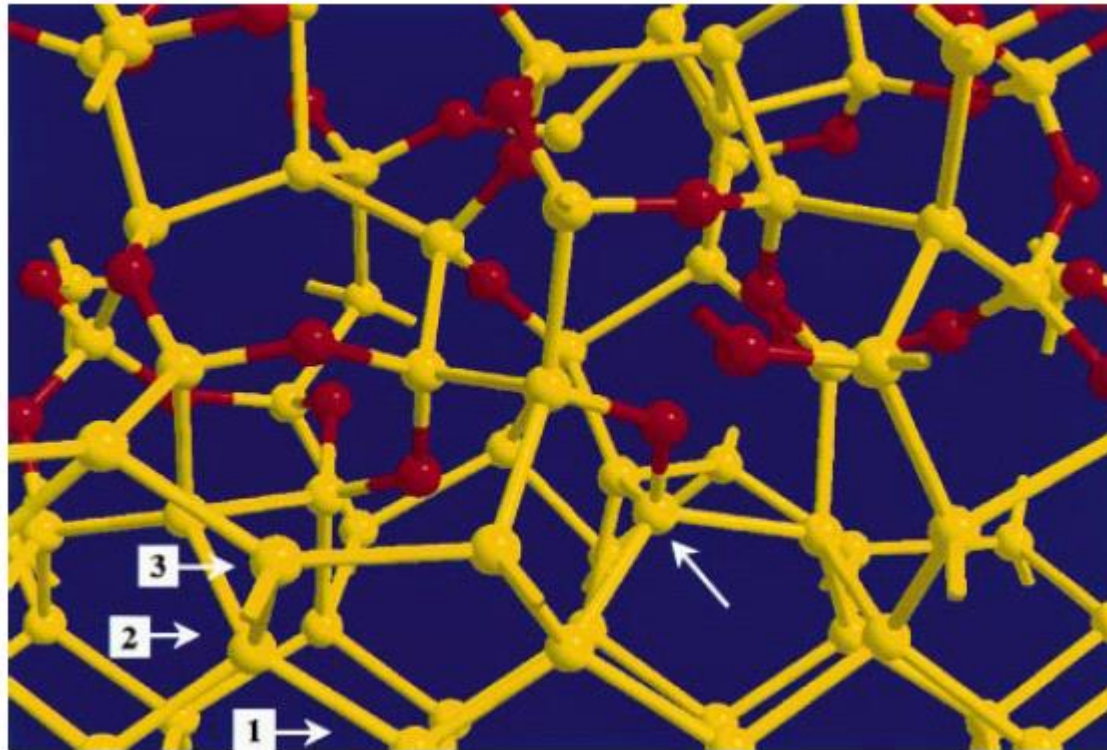
Coherent and Incoherent boundary



After W.D. Callister, Jr. Fundamental of Materials Science
And Engineering (2001)



Si/SiO₂ interface

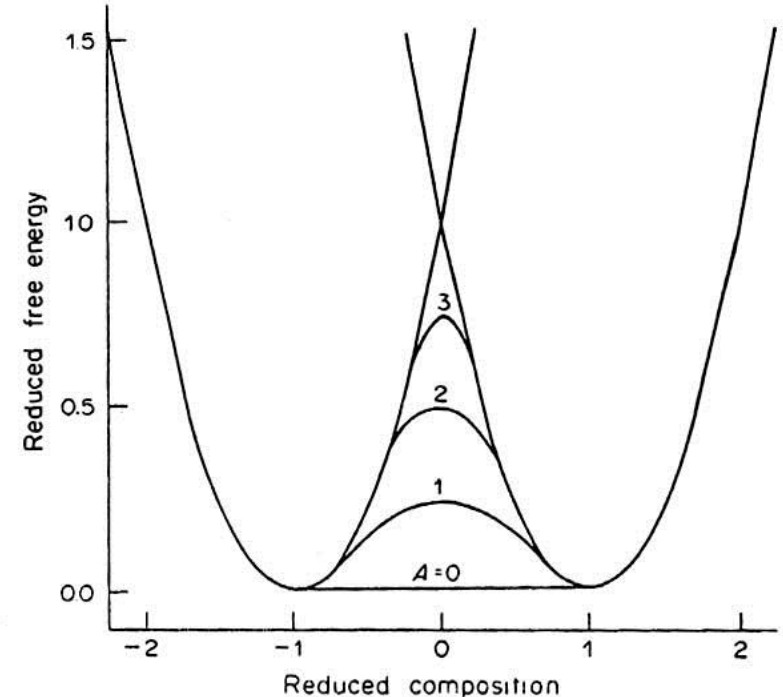
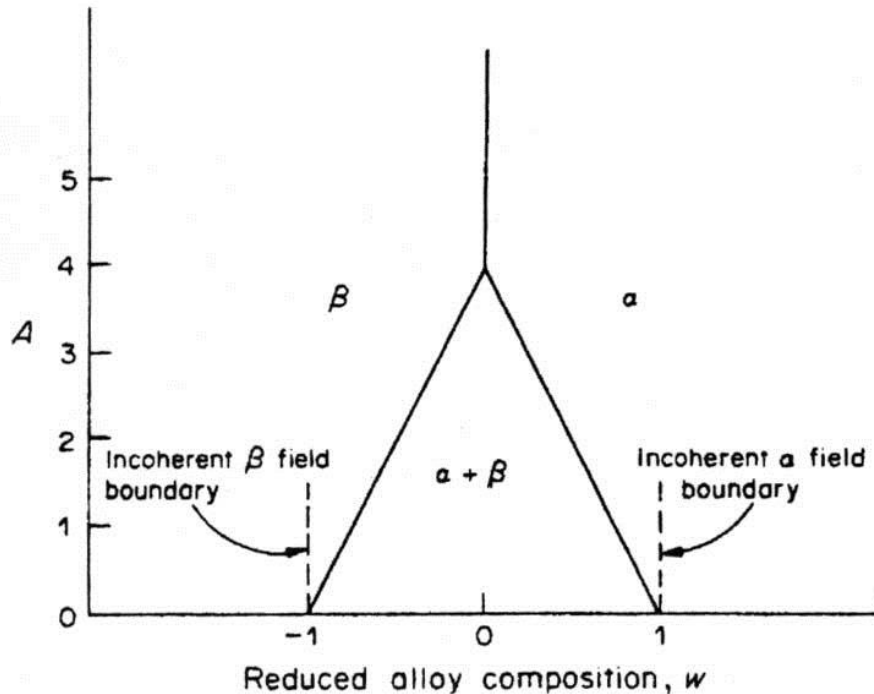


Pasquarello, Hybestsen and Car, Nature, 396 (1998) 58.

Si(001)-SiO₂, ● : Si, ● : O



Effect of coherent energy on phase diagram



Cahn and Larche, Acta Met. 32 (1984) 1915.

$$A = 4VE\varepsilon^2/[(1-\nu)F''(C_\beta^e - C_\alpha^e)^2]; \text{ Elastic energy } F_e = z(1-z)VE\varepsilon^2/(1-\nu)$$

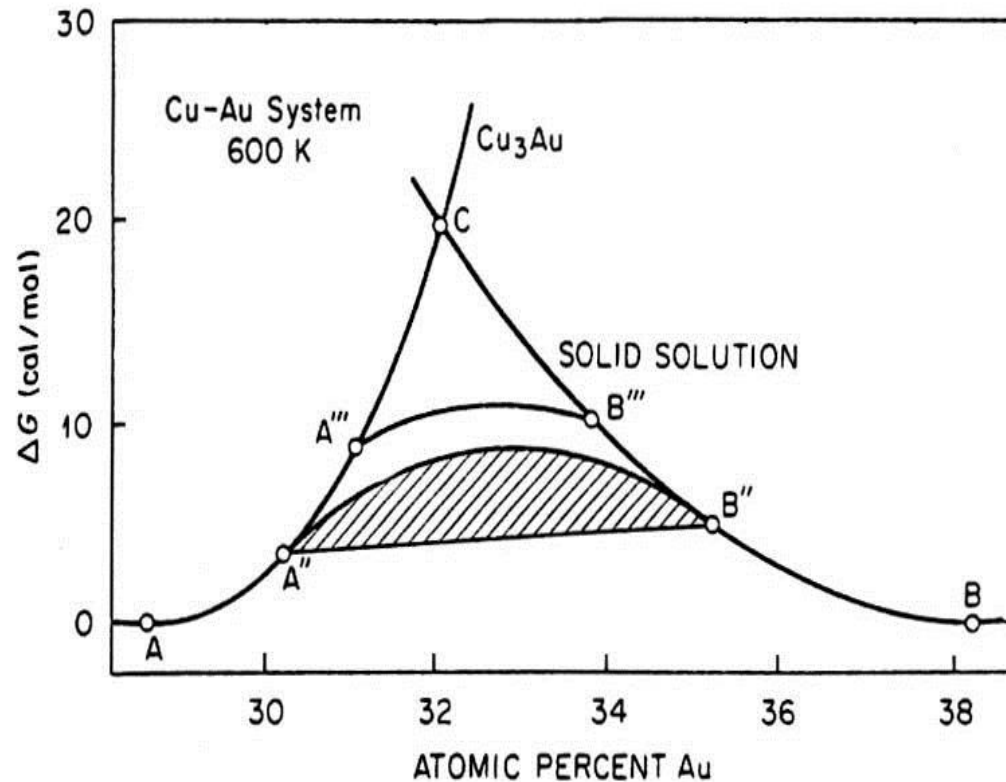
V: molar volume.

z: mol fraction of one phase

ε : linear mismatch between phases



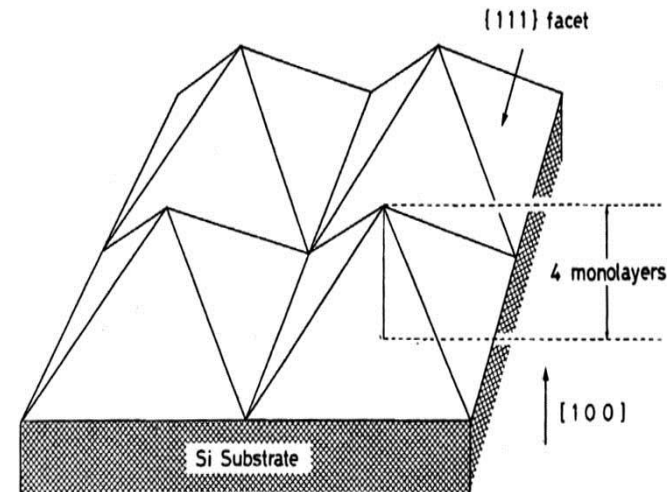
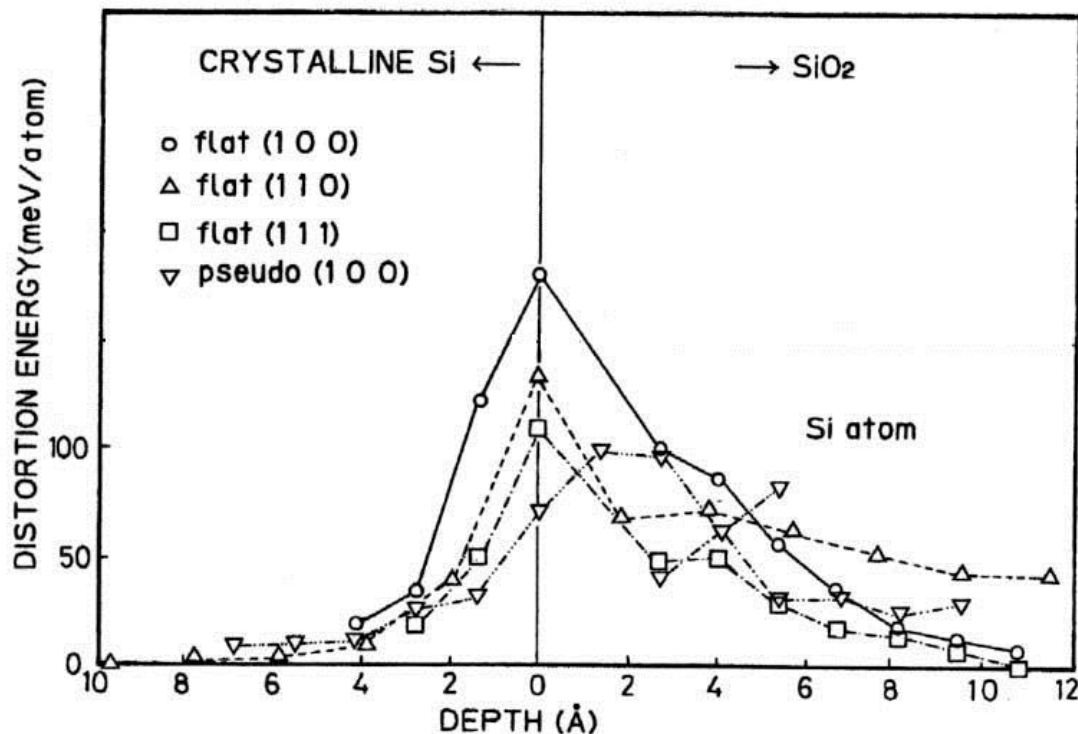
Schematic free energy diagram



Williams, R.O, Met. Trans, 11A (1980) 247.



Coherency strain energy



Ohdomari, Akatgsu, Yamakoshi and Kishimoto, J. Non-Crystalline Solids, 89 (1987) 239. $(100) > (110) > (111)$.

High energy (100) plane tends to create (111) surfaces.



Effect of oxygen deficiency on oxidant diffusion

Oxidation of Si is controlled by the transport of oxidant (O_2 or H_2O) through the oxide film being formed. Oxygen deficiency at the interface becomes diffusion traps of these oxidants, slowing down the diffusion.

Flux at the interface, J_1 .

Flux in the bulk of the film, J_2 .

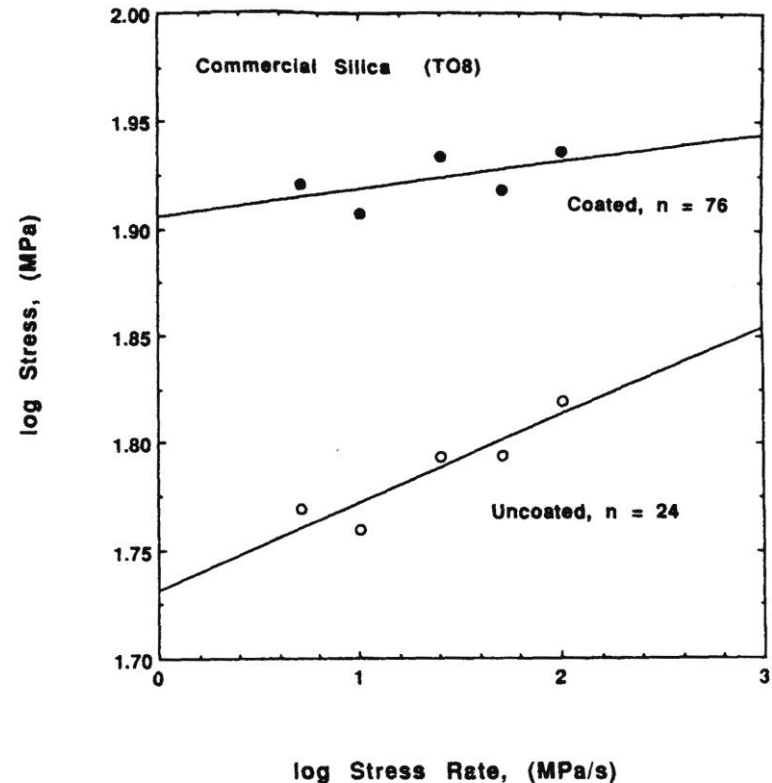
For thin films, interface flux is important \Rightarrow a linear growth.

When the film become thicker, bulk flux becomes important \Rightarrow parabolic growth.



Effect of oxygen vacancy on oxidant diffusion

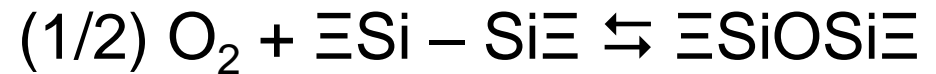
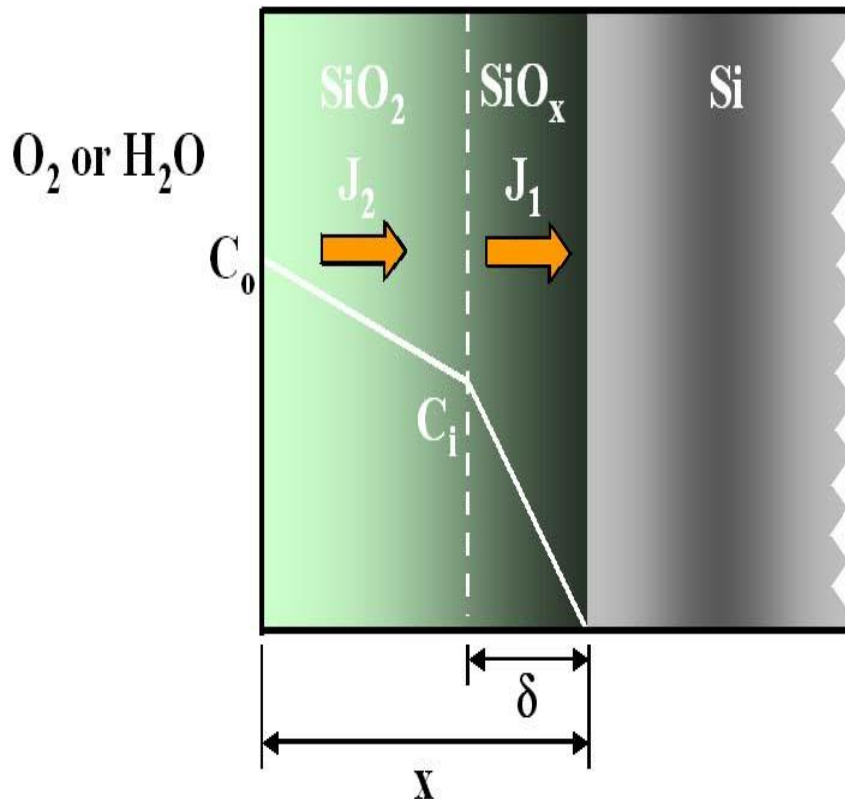
- ❑ Oxygen deficiency \approx Oxygen vacancy \approx Silicon suboxide (SiO_x , $1 < x < 2$).
- ❑ SiO_x coating on SiO_2 glass can reduce mechanical fatigue of the glass.
- ❑ Vacancy annihilation by water diffusion.
- ❑ Oxygen vacancies serve as oxidant diffusion traps and diffusion barriers.



Tomozawa, Han and Davis,
SPIE 1590 (1991) 160.

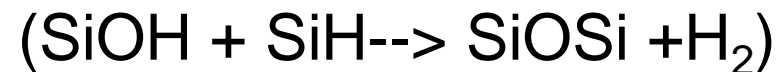
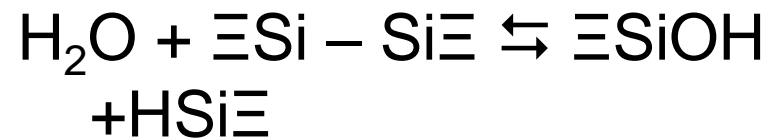


Oxygen vacancy annihilation process by oxidants



$$K_1 = C_{SiOSi} / (C_{O_2}^{1/2} C_V)$$

$$\partial C_{O_2} / \partial t - \partial C_V / \partial t = \partial [D_{O_2} \partial C / \partial x] / \partial x$$



$$K_2 = C_{SiH} C_{SiOH} / C_{H_2O} C_V$$

$$\partial C_{H_2O} / \partial t + \partial C_{SiOH} / \partial t = \partial [D_{H_2O} \partial C_{H_2O} / \partial x] / \partial x$$



Oxidant diffusion and reaction

Effective diffusion coefficient,

$$D_{\text{eff}} = D_{\text{O}_2} / [1 + (C_V / 4C_{\text{O}_2})] \approx D_{\text{O}_2} / (C_V / 4C_{\text{O}_2})$$

$$J_1 = J_2 \quad DC_0 / (x - \delta) = D_{\text{eff}} C_i / \delta = N (\partial x_0 / \partial t)$$

$$\begin{aligned} x_0^2 + 2[(D_{\text{O}_2} / D_{\text{eff}}) - 1]N \delta x_0 \\ = 2(D_{\text{O}_2} C_0 / N)t + [(2D_{\text{O}_2} / D_{\text{eff}}) - 1]\delta^2 \end{aligned}$$

Similar to Deal and Grove equation, $x^2 + Ax = B(t + \tau)$

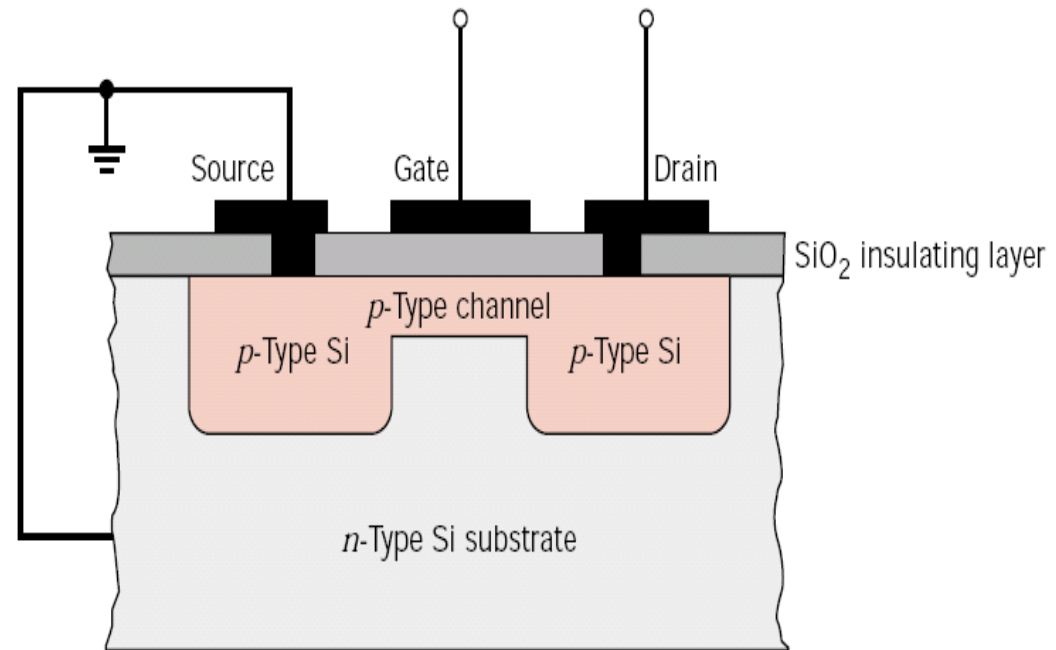
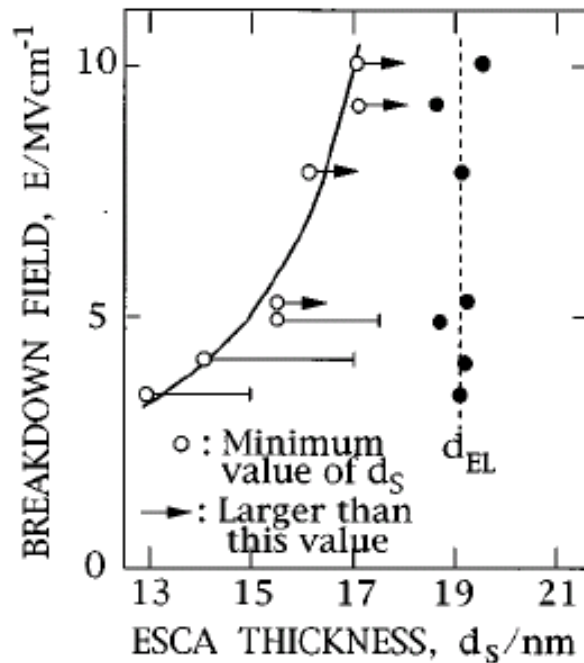
With $B = 2D_{\text{O}_2}C_{\text{O}_2}/N$

$$B/A = C_0 / \{N [(1/D_{\text{eff}}) - (1/D)] \delta\} \approx C_0 D_{\text{eff}} / (N \delta)$$

Similar equations for wet (H_2O) oxidation.



Effect on electrical properties

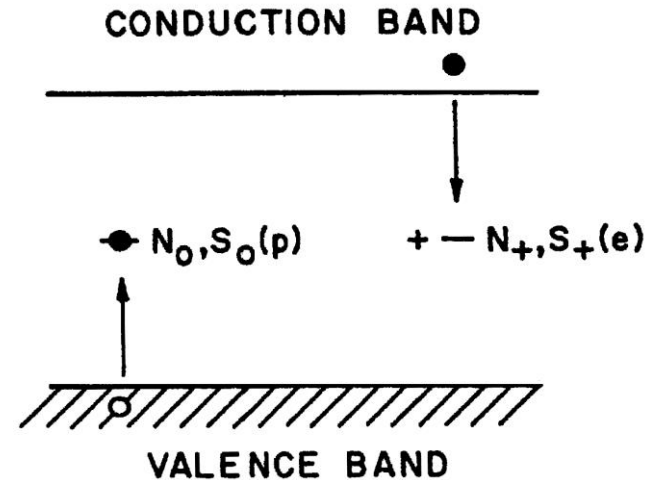
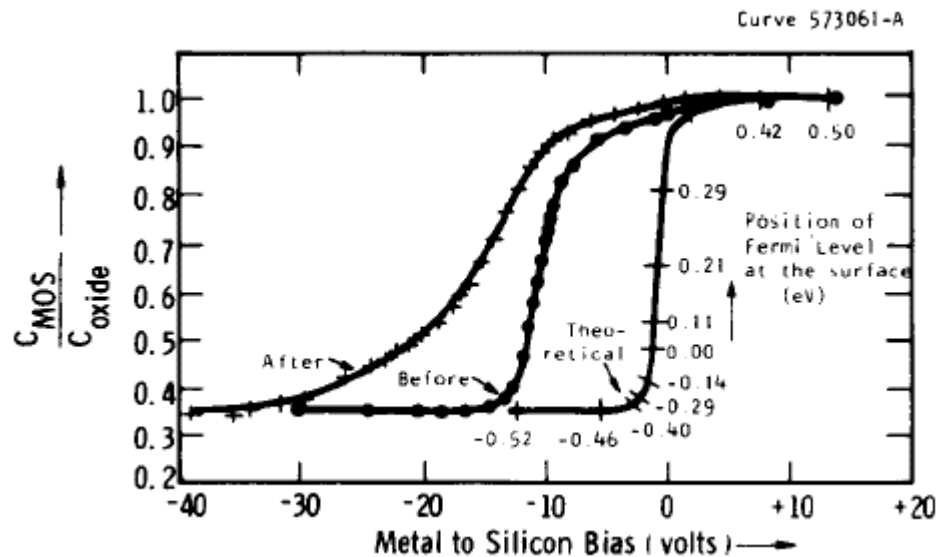


Iwata and Ishizaka, J. Appl. Phys. 79 (1996) 6653.
Si-Si is responsible-Hasegawa et al., J. Electrochem. Soc. 142 (1995) 273.

Callister, Fundamental of Materials Science and Engineering, Wiley (2001).



Effect on electrical properties

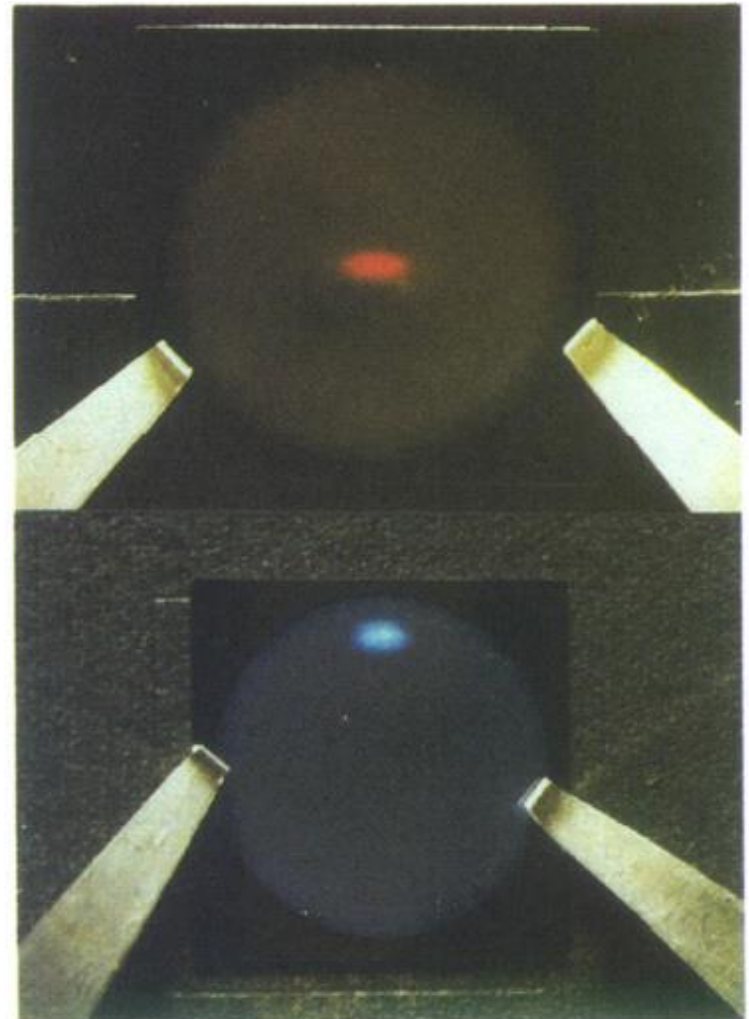
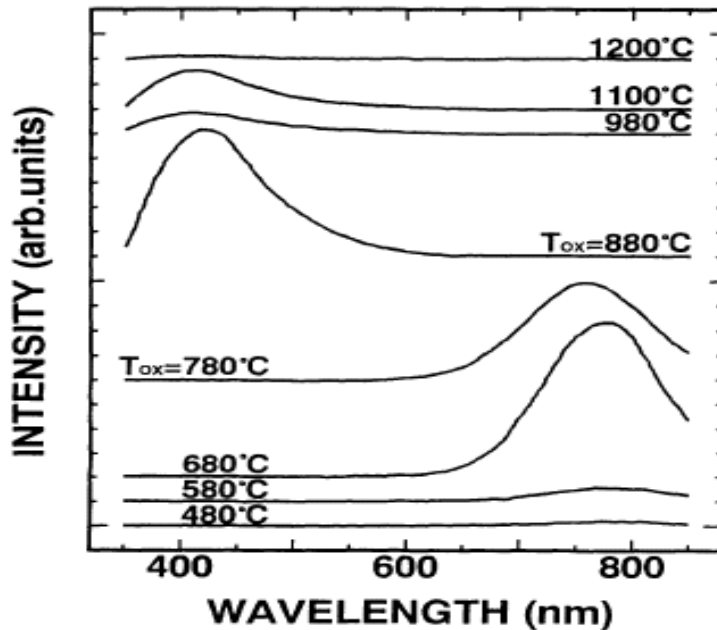


Szedon and Sandor, Appl. Phys. Letters, 6 (1965) 181.
 Before and after electron irradiation.
 Warren et al, APL, 64 (1994) 3452...
 O vacancies are the dominant hole trapping sites.

Si band diagram with hole traps.
 Woods and Williams, JAP, 47 (1976) 1082.



Effect on optical properties

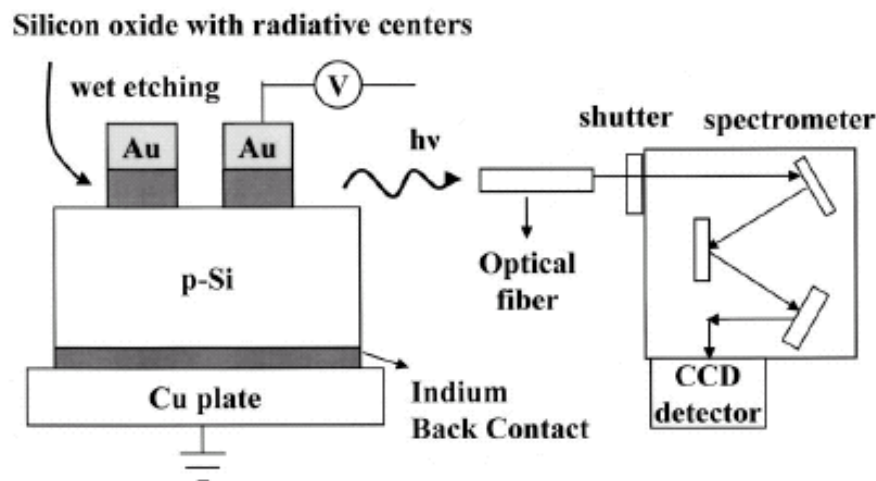


Futagi, Matsumoto and Mimura, Phys. Rev. B 49 (1994) 14732.

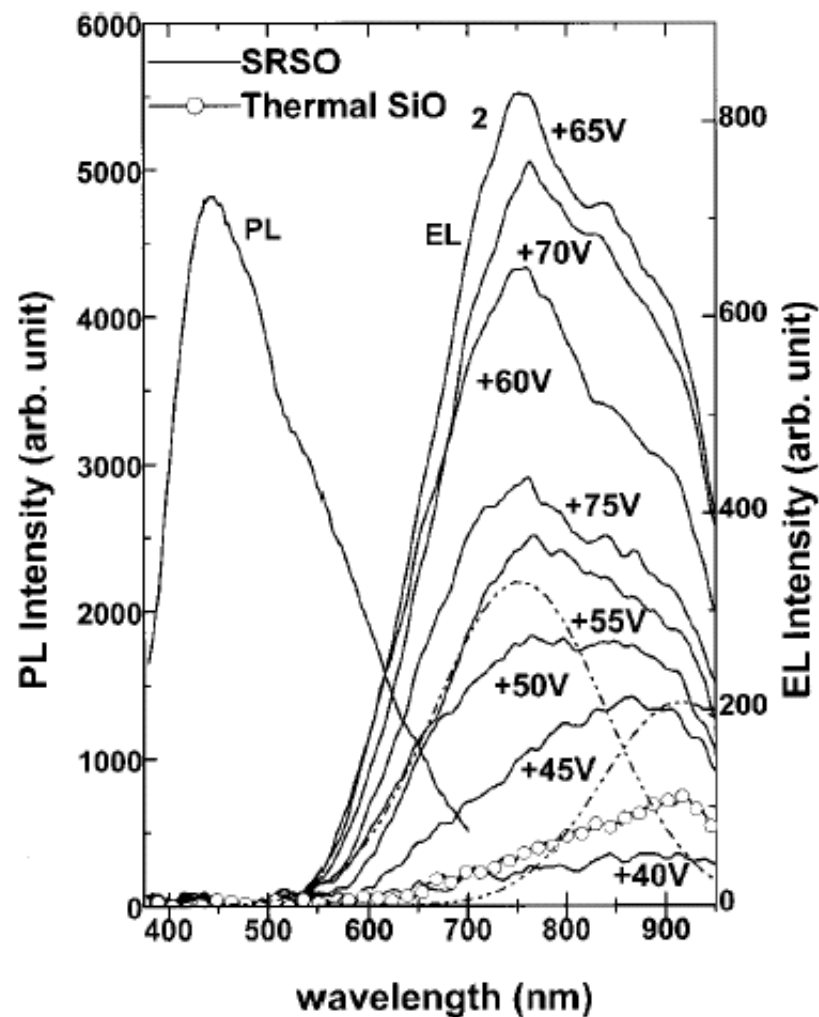
PL of oxidized porous Si. Excited with 325 nm. Oxide formed at lower temperature contains OH.



Effect on optical properties



Photoluminescence and electroluminescence, $\text{SiO}_{1.4}$ film prepared by rf sputtering, heated at 500°C . PL excited with 351 nm. PL: blue and EL:red.



Conclusions

- It is suggested that the linear growth of silica by oxidation of silicon is due to the diffusion and reaction process at the Si/SiO₂. The oxygen vacancies in SiO_{2-x} at the Si/SiO₂ interface work as the oxidant diffusion traps. The slow oxidation rate of nano-silicon and effect of different Si crystalline orientation can be attributed to different oxygen vacancy concentration at the interface.
- Oxygen vacancy influences both optical and electrical properties of Si/SiO₂ systems.



Acknowledgement

Financial support of NSF under contract, DMR-0352773.

We appreciate discussion with Professor R.H. Doremus of Rensselaer Polytechnic Institute

